

Symposium F-9

From Microstructure to Properties: Mechanisms, Microstructure, Manufacturing
Nov. 1. 2018 (Th.) 10:15~10:30



Equilibrium crystal shape of GaAs by ab-initio thermodynamics

In Won Yeu^{1,2}, Gyuseung Han^{1,2}, Cheol Seong Hwang², and Jung-Hae Choi^{1*}

¹Center for Electronic Materials, Korea Institute of Science and Technology (KIST), Korea

²Department of Materials Science and Engineering, Seoul National University, Korea

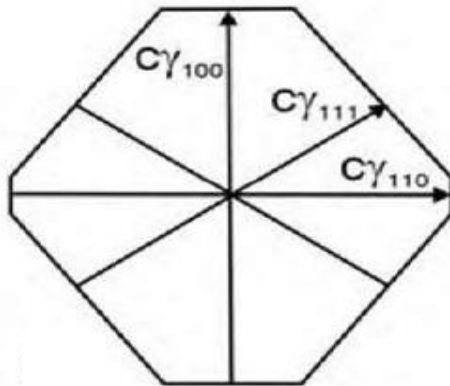
choijh@kist.re.kr

Equilibrium Crystal Shape (ECS)

Wulff construction:

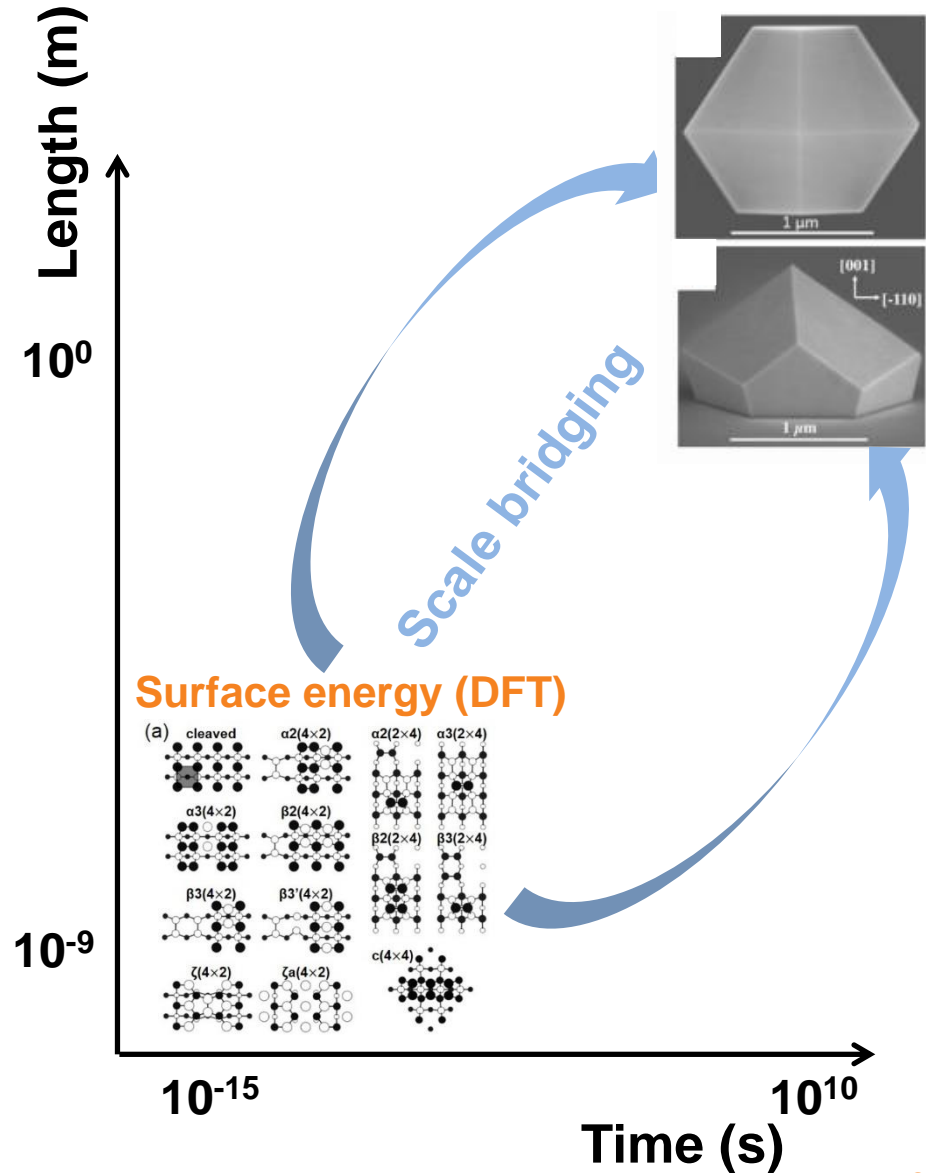
$$d^{(hkl)} \propto \gamma^{(hkl)}$$

; Minimum total surface energy



ECS(T, P)

Thermodynamics (ECS)



GaAs as a model system

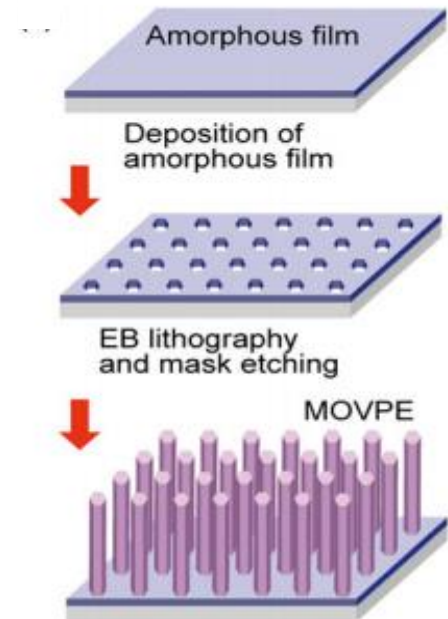
	e ⁻ mobility (cm ² /Vsec)	h ⁺ mobility (cm ² /Vsec)	Lattice constant (Å)
GaAs	8,000	400	5.65
Si	1,400	500	5.43

- Good electronic properties
- Compatibility with Si

GaAs on Si

- lattice mismatch → **Dislocation**
- Difference in thermal expansion coefficients → **Crack**
- Polar material on nonpolar → **Antiphase domain**

Selective Area Growth



Surface engineering of GaAs

Continuum
Property

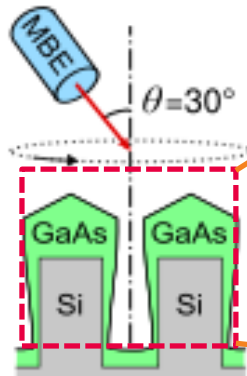
Experimental measurements

Atomic
Structure

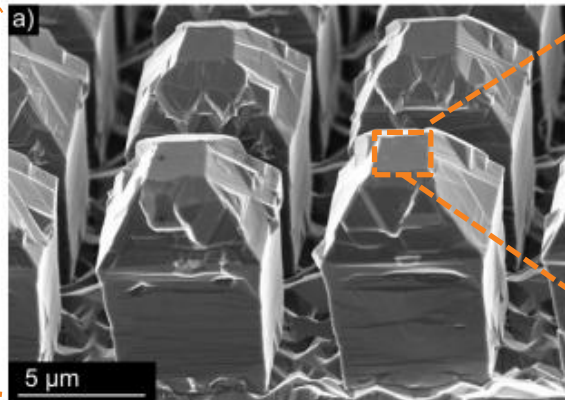
Growth Variables:
T & P

Morphology

Surface reconstruction

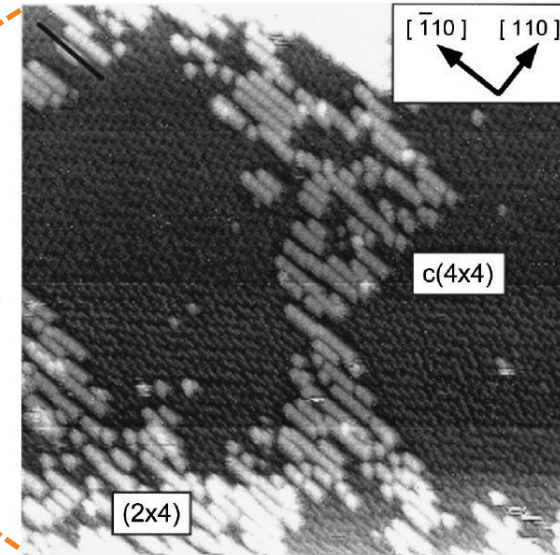


SEM



Appl. Phys. Lett. 103, 262106 (2013)

STM



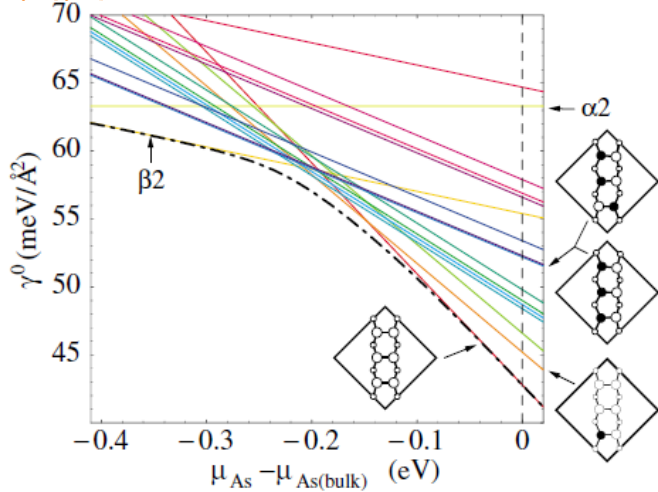
Temperature: 768 K
Pressure: $\sim 10^{-9}$ atm
Phys. Rev. B 59, 2947 (1999)

Prediction of the Reconstruction &
ECS for a given (T,P)

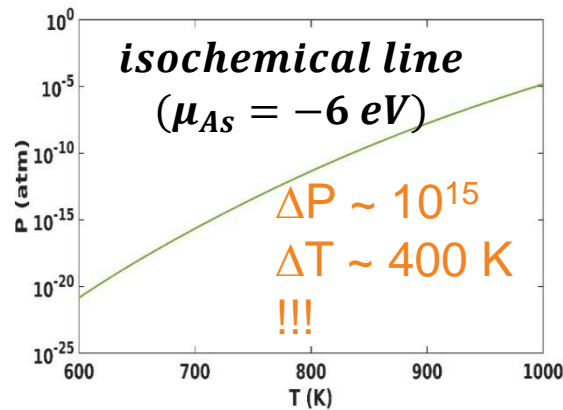
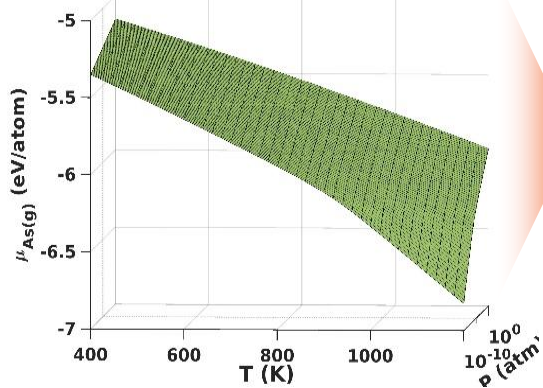
Surface energy $\gamma(\mu)$ to $\gamma(T,P)$ by DFT

Previous DFT Study

$\gamma(\mu)$ Physical review letters **93**, 146102 (2004).



$\mu_{As(g)}(T,P)$



Gap between thermodynamic variables; μ vs. (T,P)

This DFT Study

Surface energy(μ)



Surface energy(T,P)



Wulff shape(T,P)

Method 1: ab-initio thermodynamics

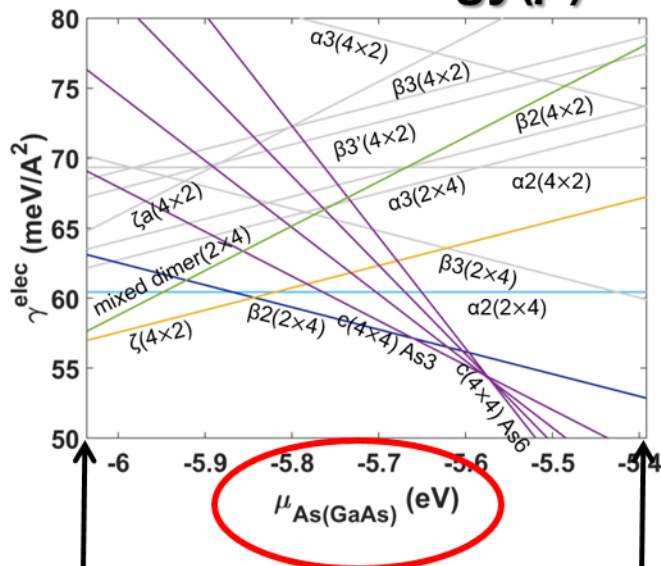
Continuum Property

DFT + Statistical calculation

Atomic Structure



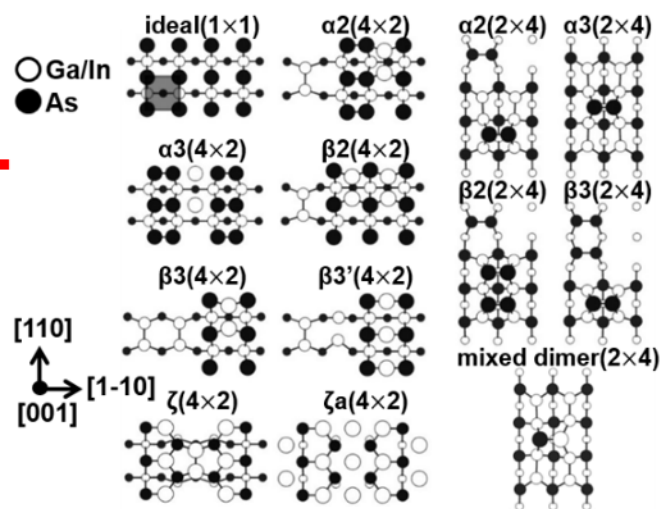
Surface energy (μ)



“Ga-rich”
High T , Low P

“As-rich”
Low T , high P

Surface reconstruction



DFT calculation

Where is (768 K, 10^{-9} atm) ??

Method 1: ab-initio thermodynamics

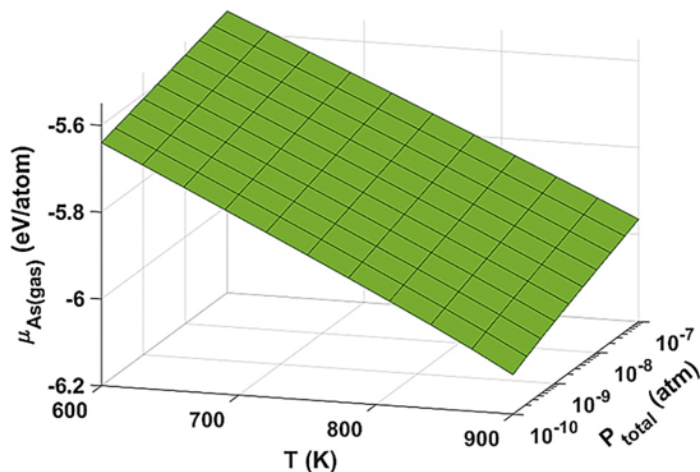
Continuum
Property

DFT + Statistical calculation

Atomic
Structure

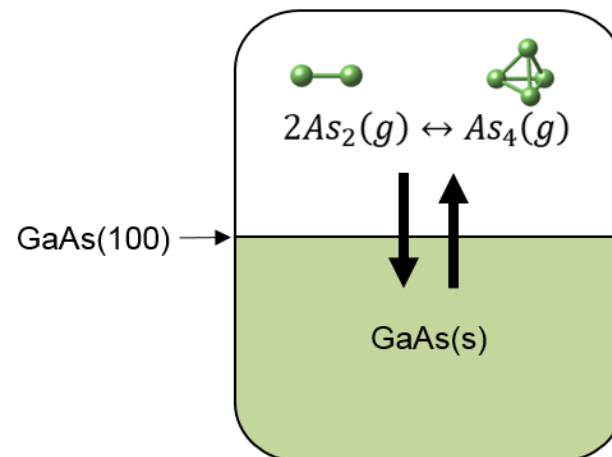
DFT calculation

$$\mu_{i(g)}(T, P) = \mu_{i(gas)}^0(T, P^0) + k_B T \ln \frac{P_{i(gas)}}{P^0}$$



Equilibrium

$$\mu_{As(g)} = \mu_{As(GaAs)}$$



x-axis: $\mu_{As(GaAs)} = \mu_{As(g)}(T, P)$!!

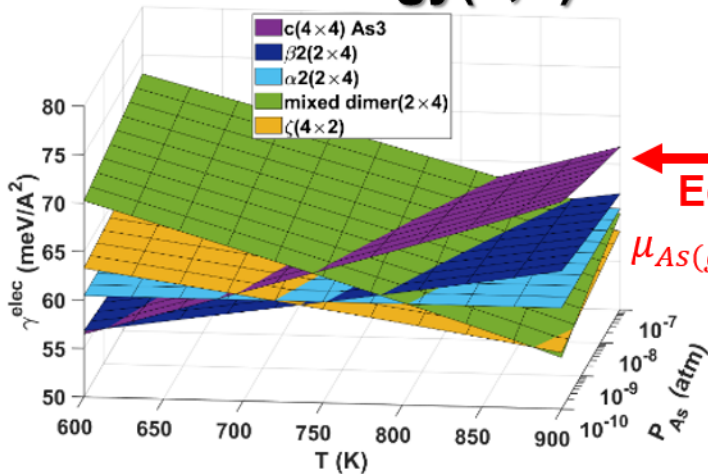
Method 1: ab-initio thermodynamics

Continuum Property

DFT + Statistical calculation

Atomic Structure

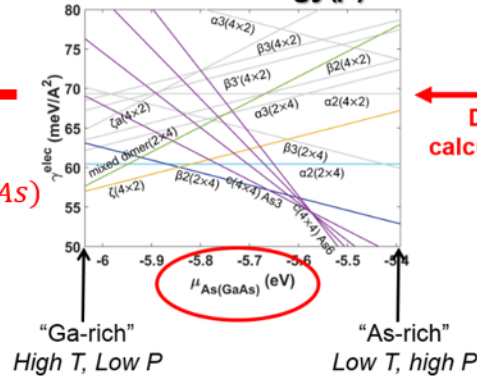
Surface energy(T,P)



Equilibrium

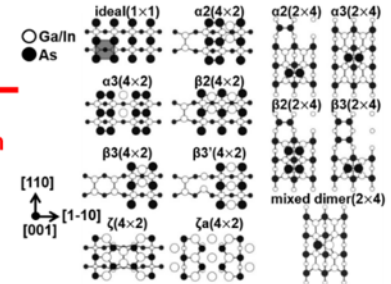
$$\mu_{As(g)} = \mu_{As(GaAs)}$$

Surface energy(μ)



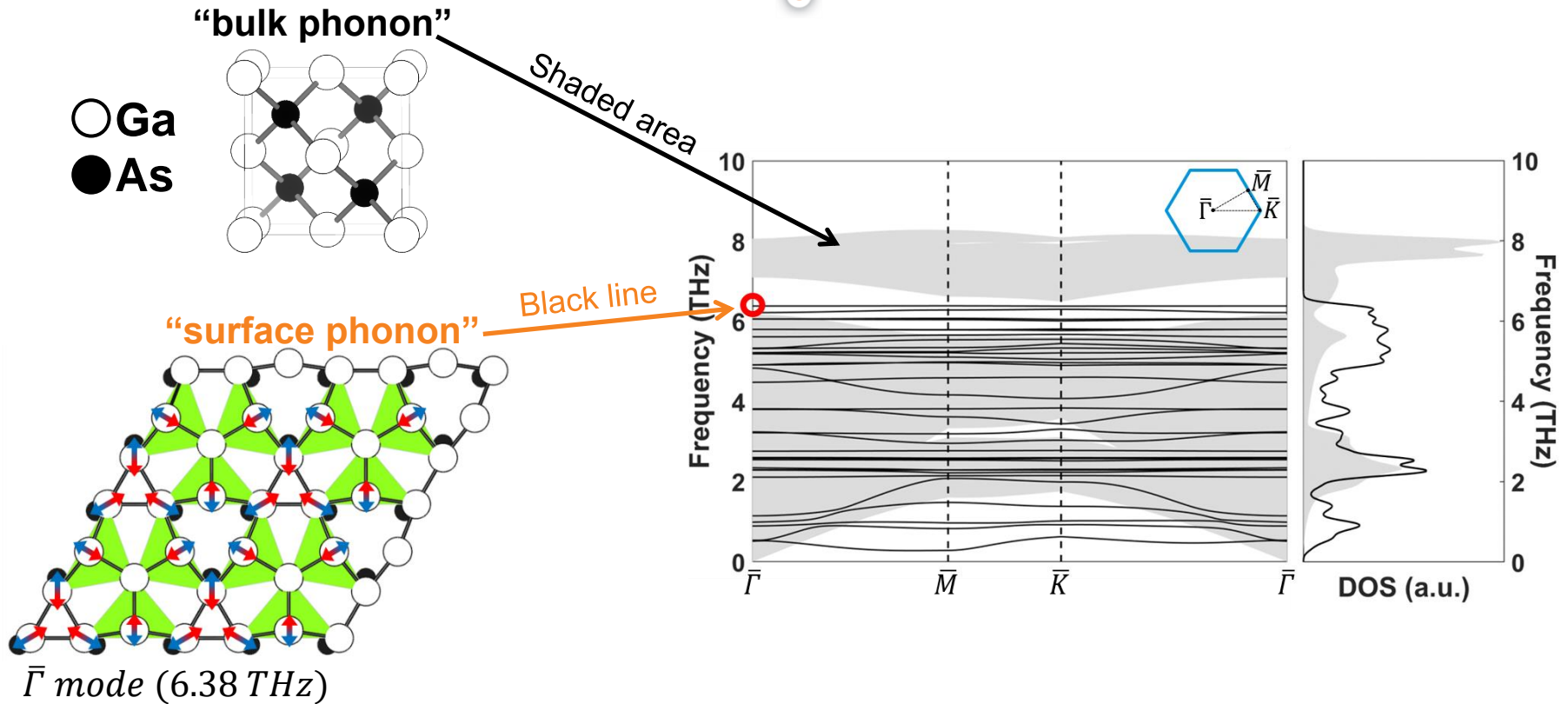
DFT calculation

Surface reconstruction



Where is (768 K, 10⁻⁹ atm) ??

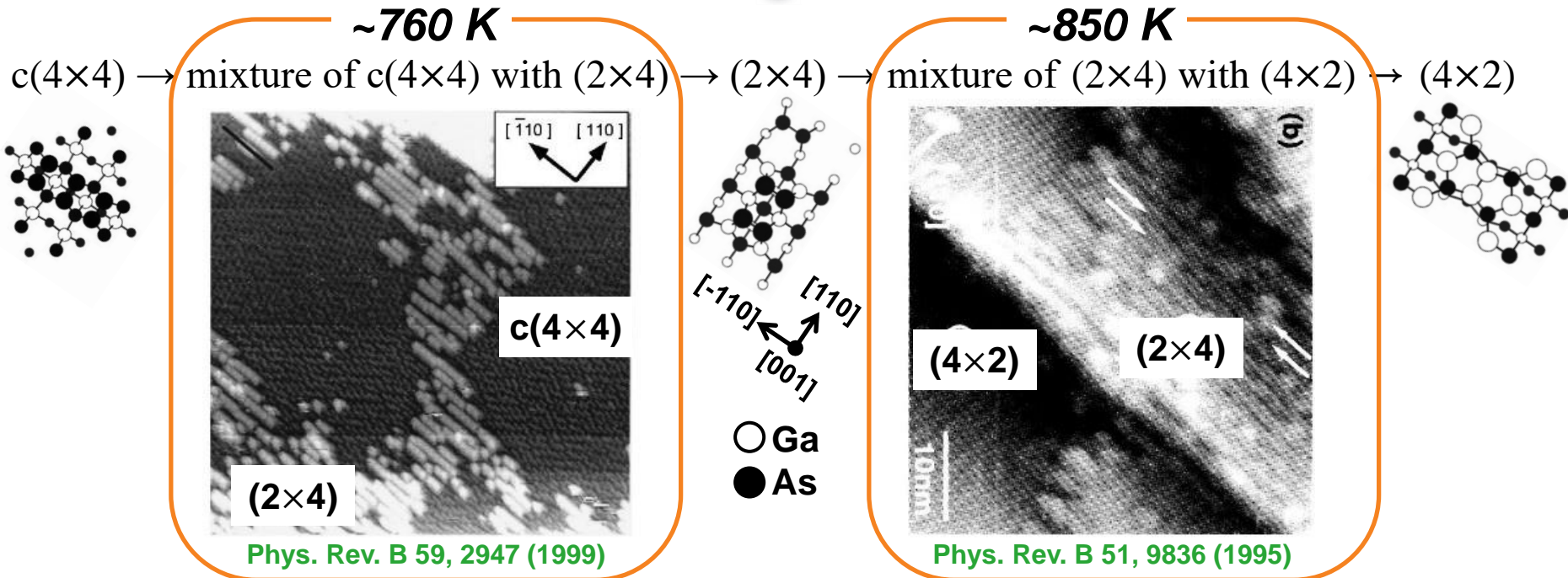
Method 2: effects of surface vibration



Surface vibration

- Frequency of surface phonon is usually lower than that of bulk phonon.
- Each reconstruction has different surface phonon frequency due to different bonding geometry and stoichiometry.

Method 3: configurational effects



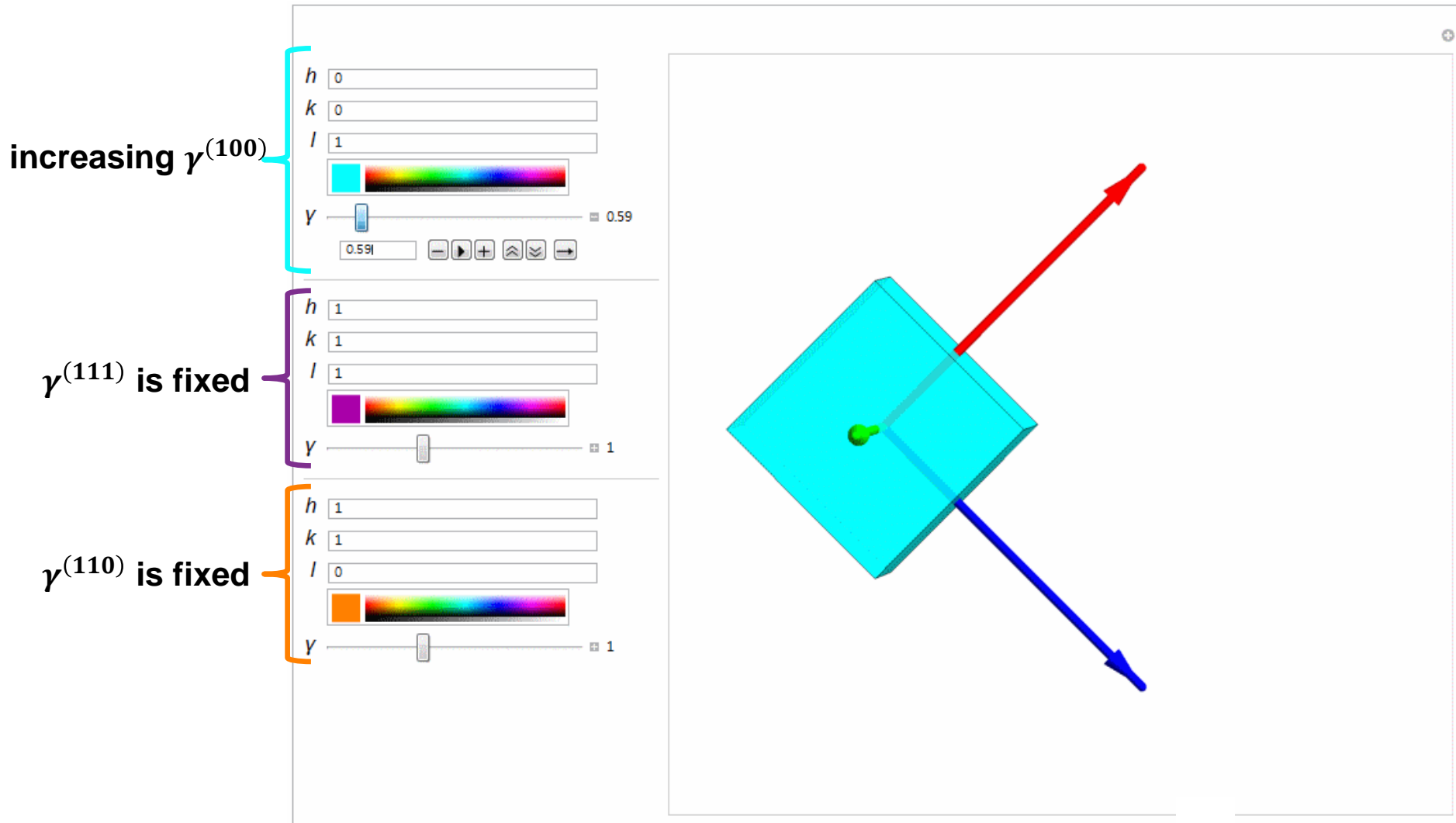
Mixed phase (coexistence)

- Real situation is not the ground state.
- Configurational entropy overcomes small energy difference at non-0 K.

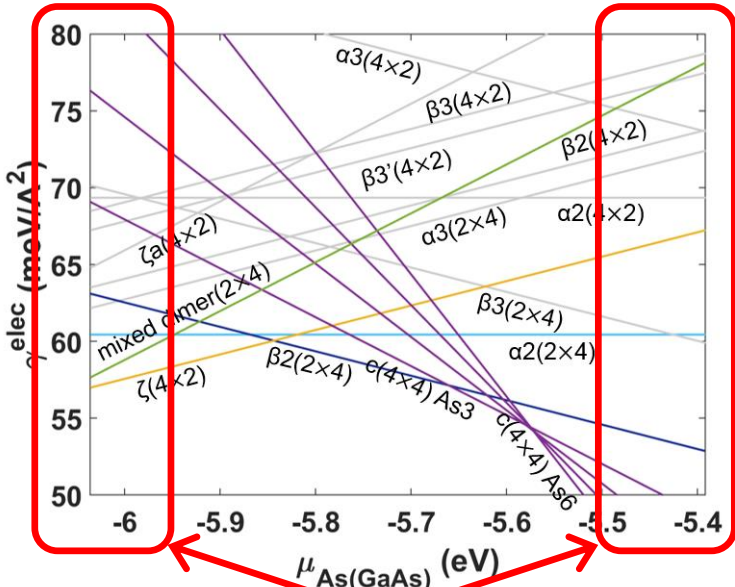
Population of reconstruction i : $c_i = \frac{Z_i}{Z}$ where $i \in \{\text{reconstructions}\}$

$$Z = \sum_i Z_i = \sum_i g_i \exp\left(-\frac{\gamma_i(T, P)A}{k_B T}\right)$$

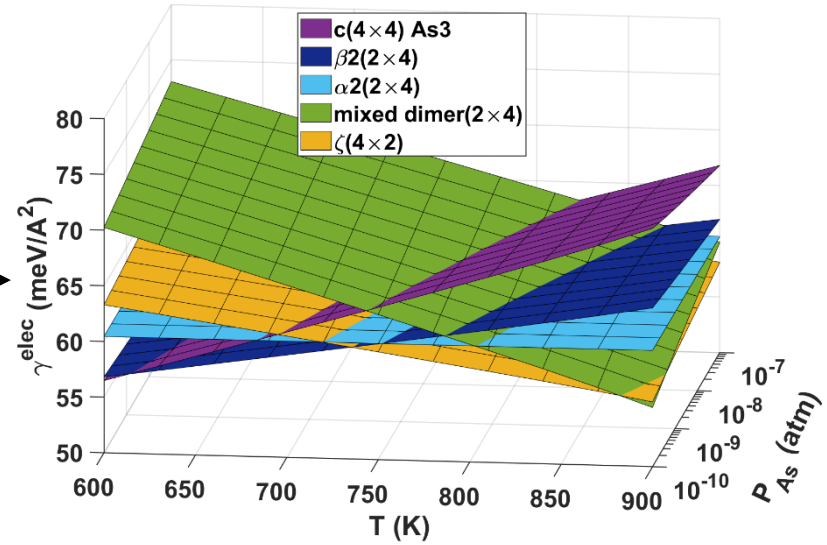
Method 4: Wulff construction



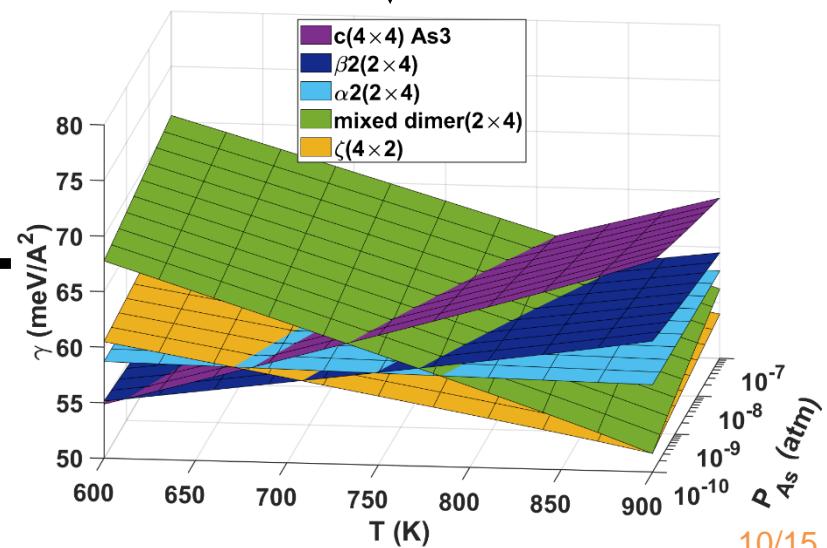
GaAs(100) phase diagram (T, P_{As})



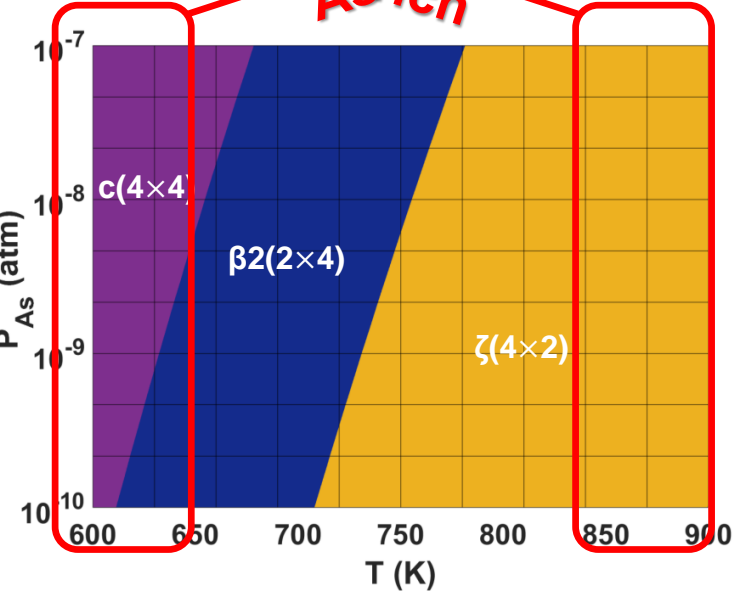
“Equilibrium”
 $\mu_{As(g)} = \mu_{As(GaAs)}$



$\gamma = \gamma^{elec} + \Delta\gamma^{vib}$

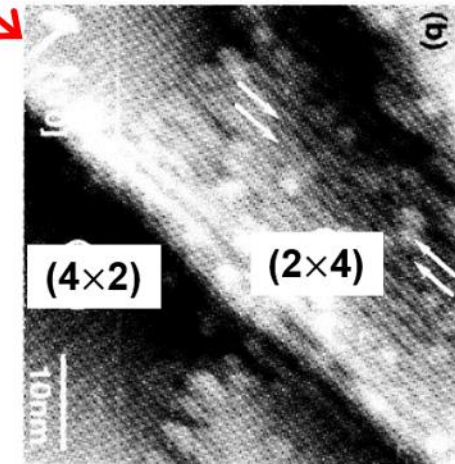
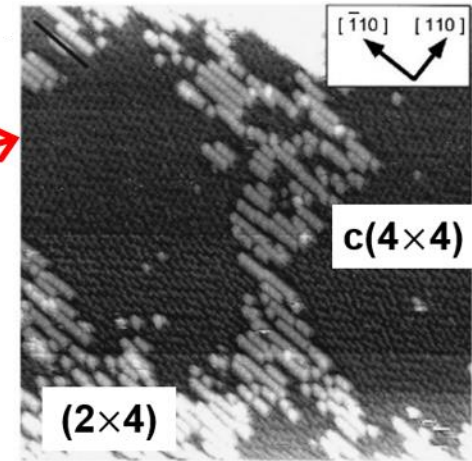
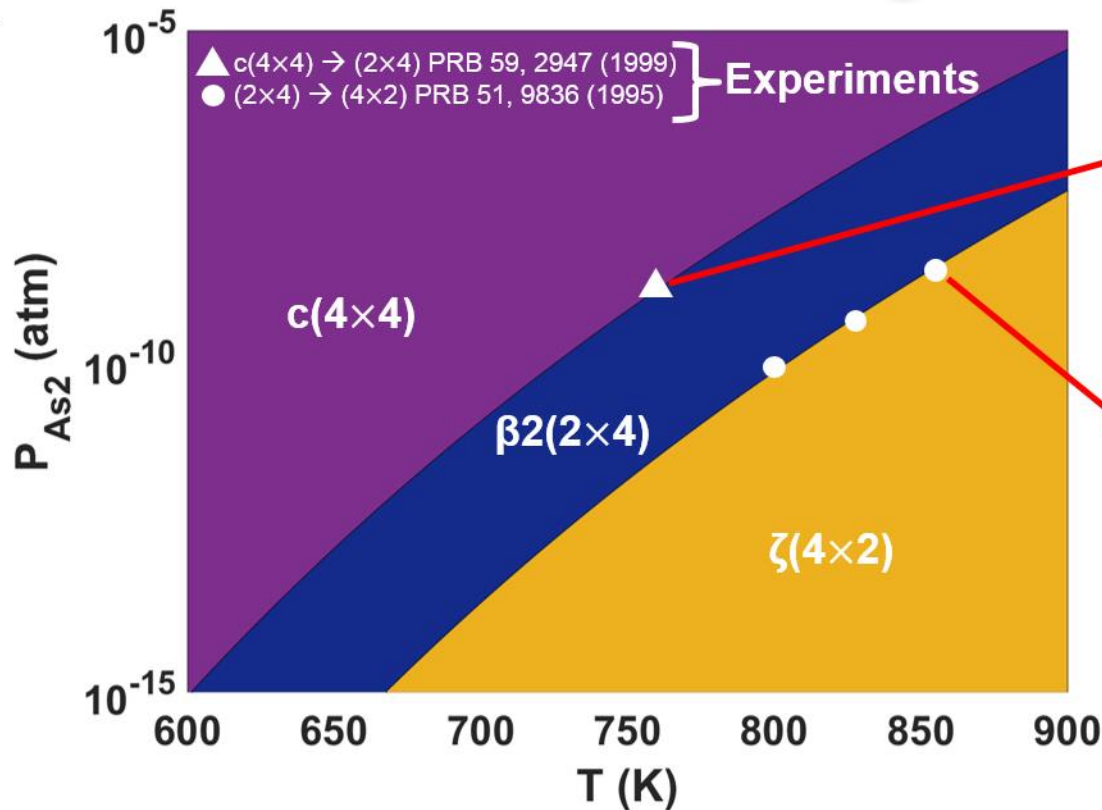


Ga-rich
As-rich



“Most stable”

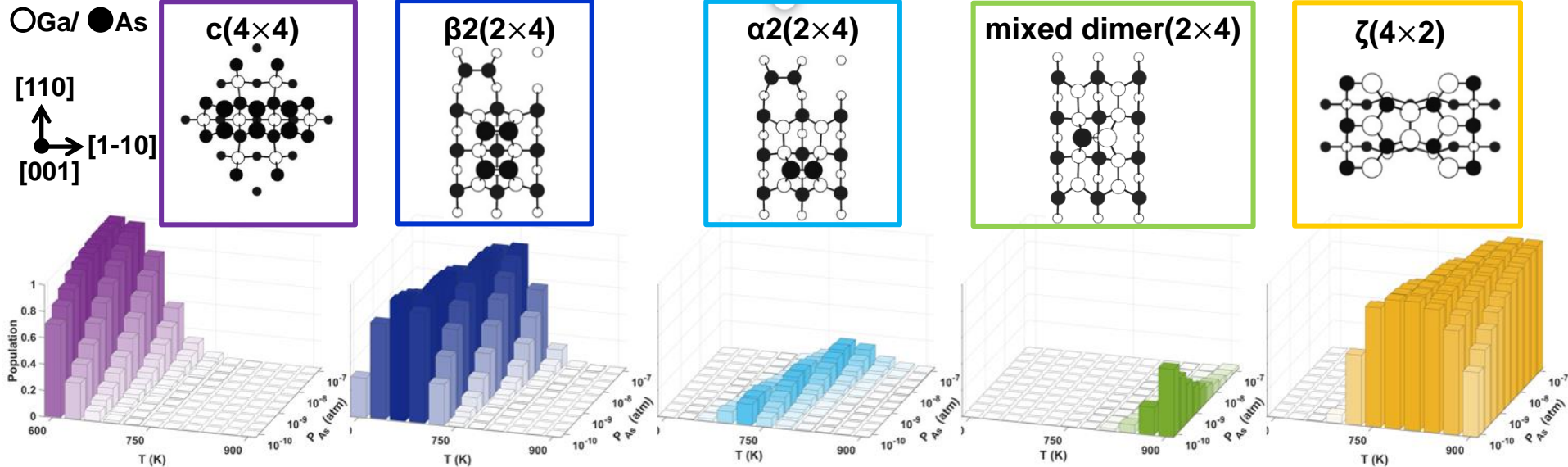
GaAs(100) surface transition (T, P_{As})



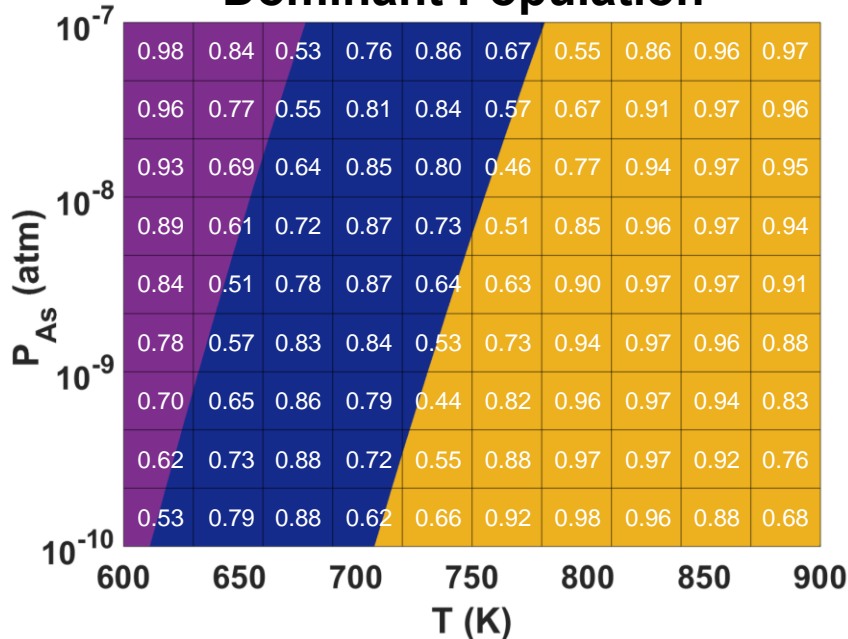
Transition Lines of GaAs(100)

- Calculated transition lines show good agreements with experimental transition (T, P) points.
- At around the transition lines, mixed phase occurs in experiments.

GaAs(100) surface population (T, P_{As})



“Dominant Population”



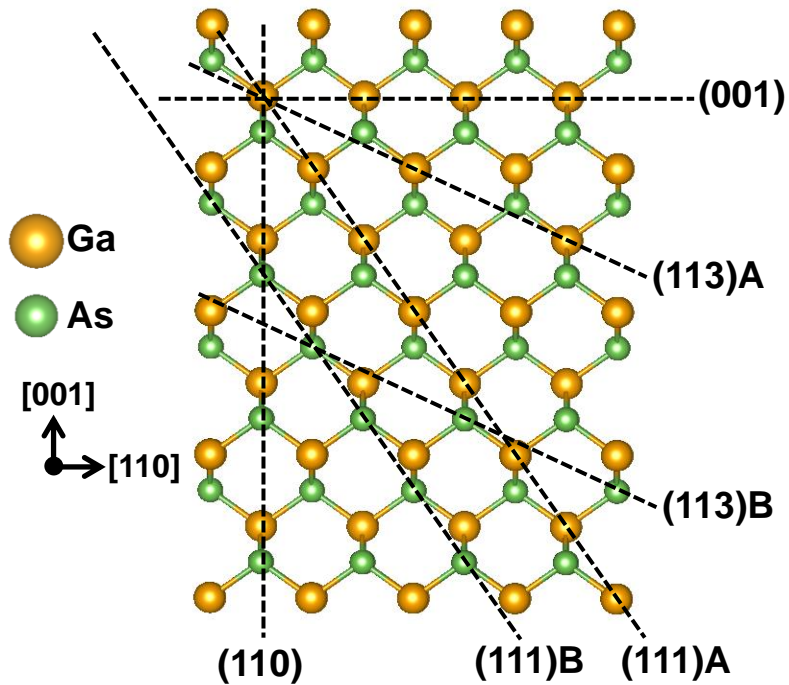
Population of reconstruction i :

$$c_i = \frac{Z_i}{Z} \text{ where } i \in \{\text{reconstructions}\}$$

$$Z = \sum_i Z_i = \sum_i g_i \exp\left(-\frac{\gamma_i(T, P)A}{k_B T}\right)$$

Various surfaces of GaAs

GaAs (Zinc Blende structure)

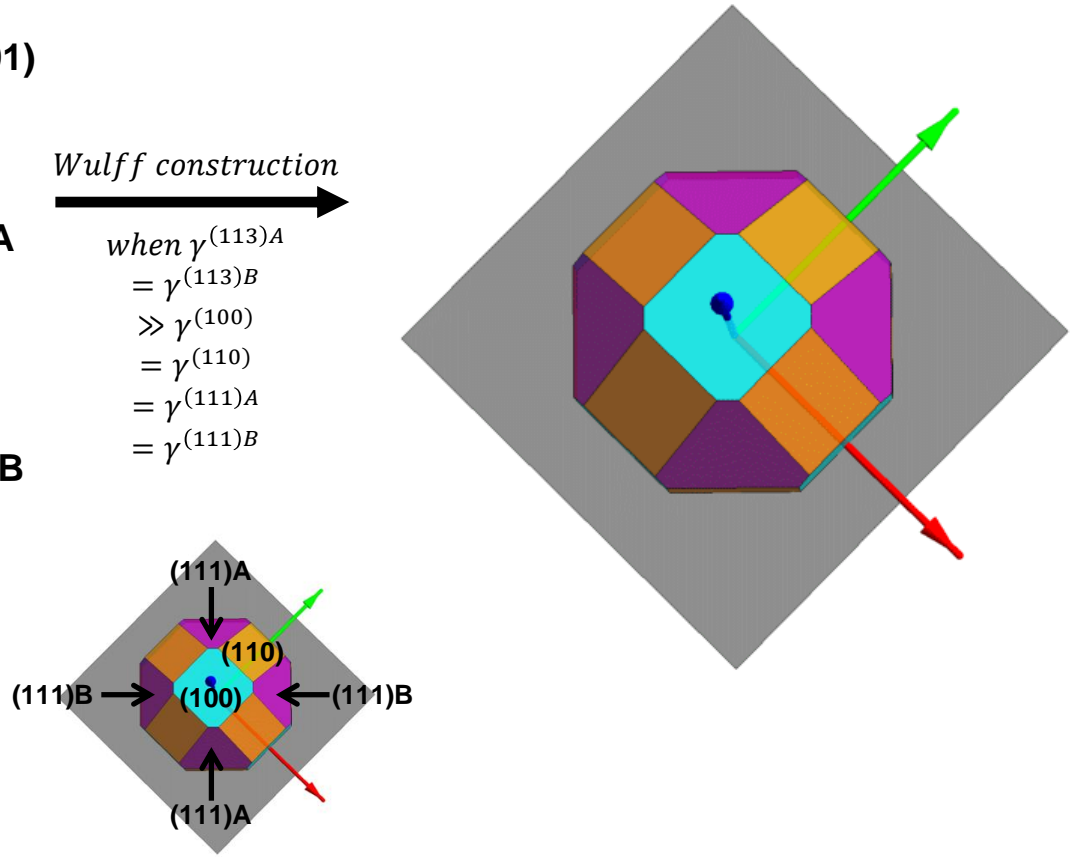


Wulff construction



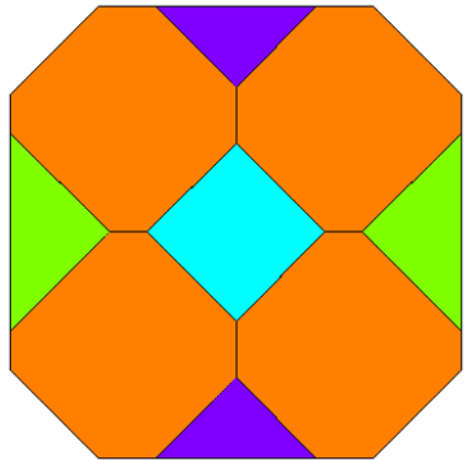
$$\begin{aligned}
 &\text{when } \gamma^{(113)A} \\
 &= \gamma^{(113)B} \\
 &\gg \gamma^{(100)} \\
 &= \gamma^{(110)} \\
 &= \gamma^{(111)A} \\
 &= \gamma^{(111)B}
 \end{aligned}$$

Wulff shape



Wulff construction: $d^{(hkl)} \propto \gamma^{(hkl)}$

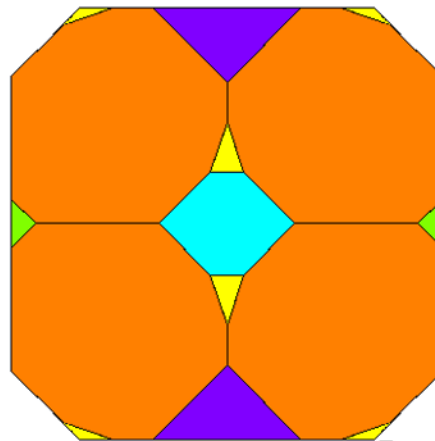
Equilibrium Crystal Shape (T, P_{As})



T: 1023 K, P: 10^{-5} atm

$$\mu_{As(g)} = -6.04$$

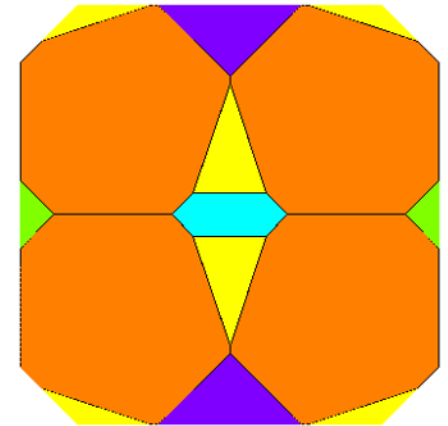
4



T: 923 K, P: 10^{-5} atm

$$\mu_{As(g)} = -5.91$$

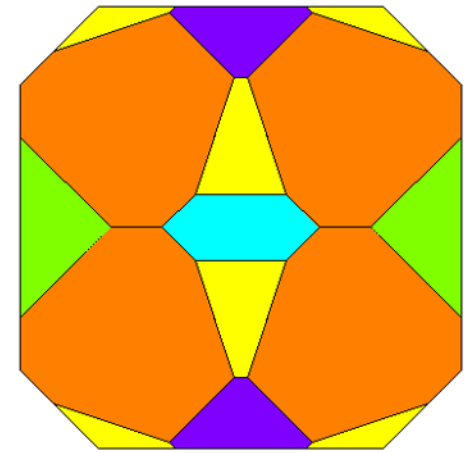
3



T: 823 K, P: 10^{-5} atm

$$\mu_{As(g)} = -5.77$$

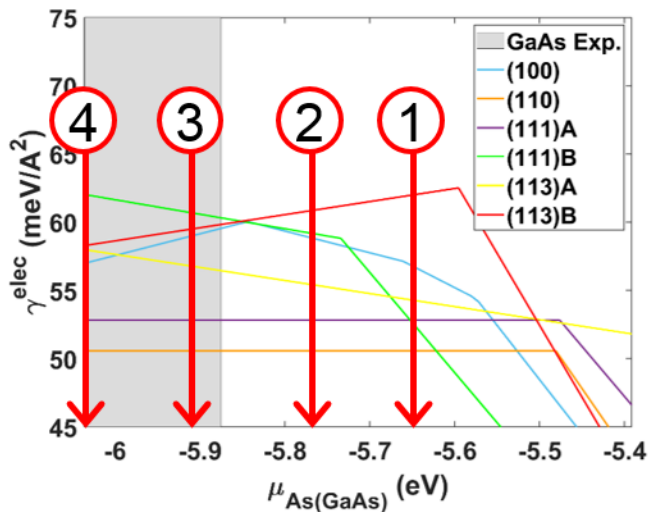
2



T: 723 K, P: 10^{-5} atm

$$\mu_{As(g)} = -5.65$$

1

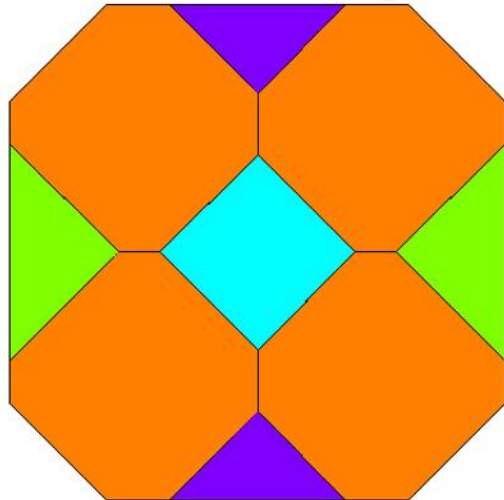


[110]
↑
[001] → [1-10]

- (100) × 6
- (110) × 12
- (111)A × 4
- (111)B × 4
- (113)A × 12
- (113)B × 12

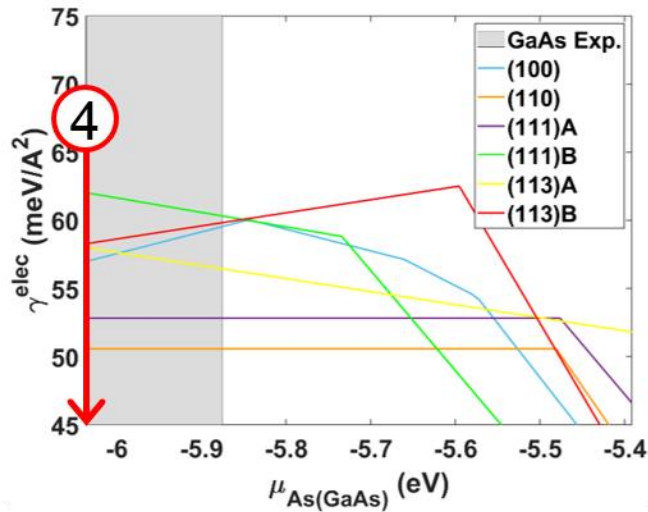
Equilibrium Crystal Shape (T, P_{As})

Wulff shape

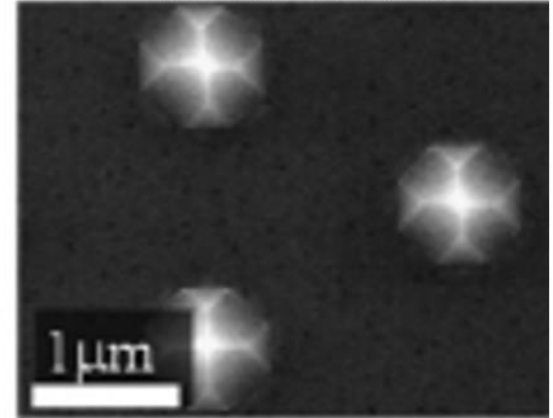


T: 1023 K, P: 10^{-5} atm

$$\mu_{As(g)} = -6.04$$



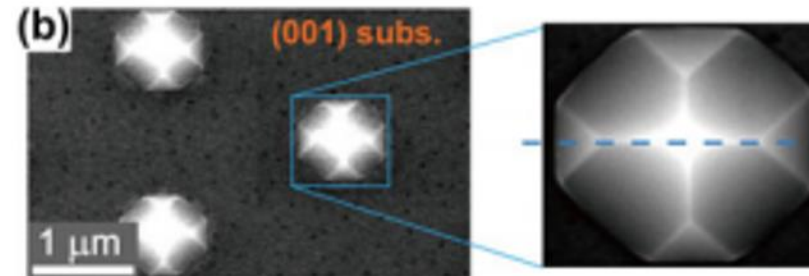
SEM of GaAs



MOCVD(TMGa & AsH₃)

T: 1023 K

J. Cryst. Growth, 298, 616 (2007)



MOCVD(TMGa & TBAs/AsH₃)

T: 973~1023 K

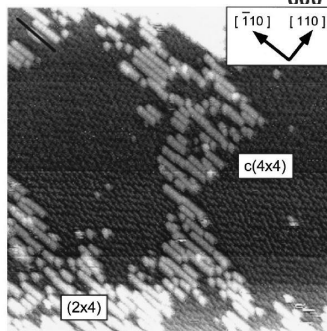
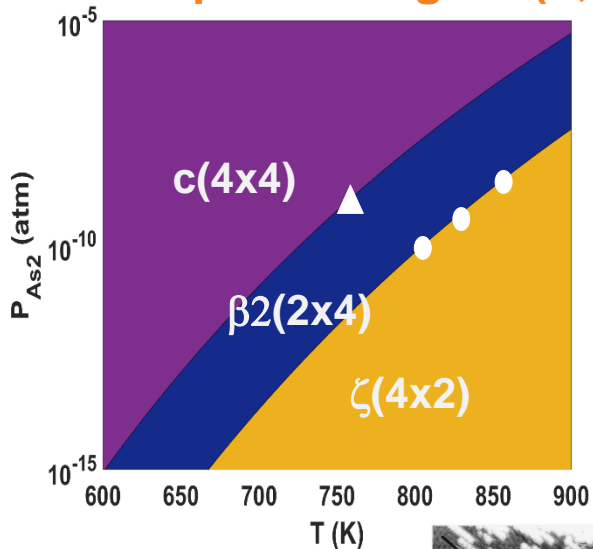
J. Mater. Res., 26, 2127 (2011)

Summary

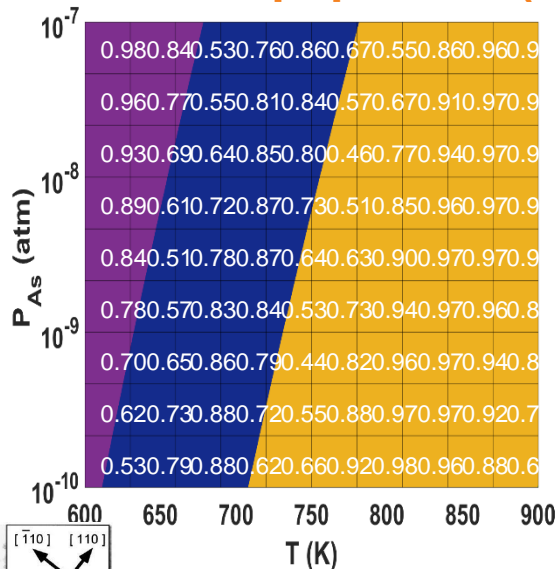
Surface energy(T,P)

Wulff shape(T,P)

Surface phase diagram(T,P)



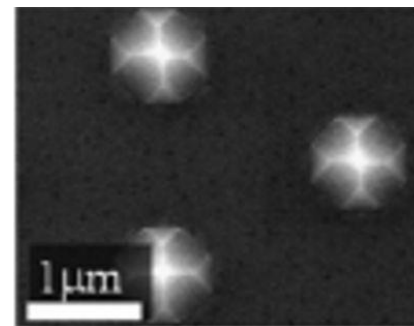
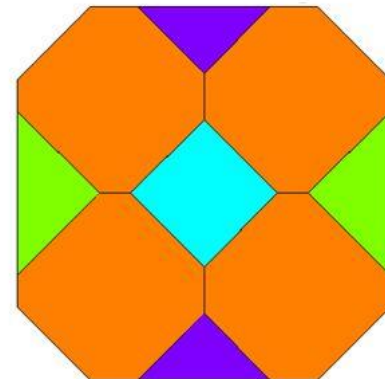
Surface population(T,P)



ECS(T,P)

$[110]$
 $[001]$ $[1-10]$

- (100)
- (110)
- (111)A
- (111)B
- (113)A
- (113)B



→ can be applicable to other systems.